Tunable altermagnetism via interchain engineering in parallel-assembled atomic chains

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Altermagnetism has recently drawn considerable attention in three- and two-dimensional materials. Here we extend this concept to quasi-one-dimensional (Q1D) monolayers assembled from single-atomic magnetic chains. Through systematically examining nine types of structures, two stacking orders, intra- and interchain magnetic couplings, we identify four out of 30 promising structural prototypes for hosting altermagnetism, which yields 192 potential monolayer materials. We further confirm eight thermodynamically stable Q1D monolayers via high-throughput calculations. Using symmetry analysis and first-principles calculations, we find that the existence of altermagnetism is determined by the type of interchain magnetic coupling and predict three intrinsic altermagnets, CrBr₃, CrCl₃, rel₃, and MnBr₃, due to their ferromagnetic interchain couplings and five extrinsic ones, CrF₃, CrCl₃, CrI₃, FeCl₃, and CoTe₃, ascribed to their neglectable or antiferromagnetic interchain couplings. Moreover, the interchain magnetic coupling here is highly tunable by manipulating the interchain spacing, leading to experimentally feasible transitions between altermagnetic and nodal-line semiconducting states. In addition, applying external electric fields can further modulate the spin splitting. Our findings establish a highly tunable family of Q1D altermagnets, offering fundamental insights into the intricate relationship between geometry, electronic structure, and magnetism. These discoveries hold significant promises for experimental realization and future spintronic applications.

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Recent advancements in magnetic and spin groups have uncovered many unconventional antiferromagnetic (AFM) materials exhibiting zero net magnetization in real space and spin-split bands with crystal symmetry paired spinmomentum locking in momentum space even without spinorbit coupling [1-6] and can be described by the spin group [7-12], a collinear subset of which is termed altermagnets [13–16]. The magnetic unit cell of an altermagnet comprises an even number of magnetic atoms that form two magnetic sublattices with antiparallel spins, which are related by crystallographic rotation or mirror operations, rather than inversion or translation symmetry operations, leading to the breaking of joint parity and time (P-T) symmetry or time reversal and translation symmetry $(T\tau)$ [13,14]. Altermagnets combine the advantages of both ferromagnets (spin-polarized bands) and antiferromagnets (zero net magnetization), thereby mitigating stray magnetic field-related issues. As a result, they exhibit a range of novel properties [4,6,17-33], such as the giant piezomagnetism [6], piezo-Hall effect [34], noncollinear spin current [6,17], and anomalous Hall effect

[29,30]. To date, more than 200 intrinsic three-dimensional altermagnetic materials have been theoretically predicted, with some experimentally confirmed [15,16]. However, the prediction of intrinsic two-dimensional (2D) altermagnets remains comparatively limited [6,17–19,35–38], primarily ascribed to stringent symmetry constraints intrinsic to 2D materials. Though these inherent geometric limitations can be transcended through artificial engineering [39–47], including the introduction of twisting angle [39] and application of external electric fields [47], the exploration of their tunability is still in its infancy.

One-dimensional (1D) materials have largely been overlooked as potential altermagnets, even including artificially engineered structures, due to their restrictive symmetry operations. Recent experimental breakthroughs have demonstrated the synthesis of single-atomic magnetic chains of CrCl₃ on the surface [48] or encapsulated within carbon nanotubes [49,50] and their self-assembled ribbons [48]. In these ribbons, while strong covalent bonds connect atoms within individual chains, van der Waals (vdW) interactions predominantly govern interchain couplings. This anisotropic nature of interactions highlights the feasibility of artificially arranging these chains into quasi-one-dimensional (Q1D) monolayers with tunable interchain spacing and stacking order. The highly adjustable interchain geometry differentiates these Q1D monolayers from conventional 2D monolayers, enabling precise control

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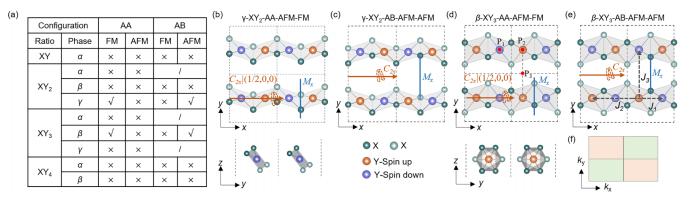


FIG. 1. (a) Summary of the emergence of altermagnetism in 1D magnetic chains with different stoichiometric ratios under AA and AB stacking configurations. FM and AFM represent interchain magnetic ordering. The symbol "×" indicates the absence of altermagnetism, while " $\sqrt{}$ " signifies its emergence. The symbol "/" represents the absence of the AB stacking configuration. Top (upper panel) and side (lower panel) views of the AA-stacked (b) and AB-stacked (c) γ -phase XY_2 (X = transition metal, Y = chalcogen/halogen atom) and AA-stacked (d) and AB-stacked (e) β -phase XY_3 monolayers. Orange arrows and blue lines illustrate symmetry operations C_{2x} and M_x that connect the sublattices with opposite spins. Red dots P₁ to P₃ marked in panel (d) indicate structural inversion centers. Orange and blue spheres represent magnetic atoms with up and down majority spins, respectively. J_1 , J_2 , and J_3 marked in panel (e) represent spin-exchange parameters for the nearest, second-nearest, and third-nearest neighbors, respectively. (f) Diagram of spin-splitting symmetry in the Brillouin zone.

over their interchain magnetic couplings [51-53]. Such a controllable interchain magnetism can be strategically tuned to meet the symmetry requirements necessary for altermagnetism, presenting distinctive opportunities to extend the scope of monolayer altermagnets and to explore novel altermagnetic properties.

Here we employed a self-assembly strategy to construct Q1D monolayer configurations from XY_n 1D single-atomic magnetic chains. By systematically exploring chemical stoichiometric ratios, structural phases, stacking orders, and both interchain and intrachain magnetic couplings, we identified four altermagnetic structural prototypes among 30 candidates. Expanding on these prototypes, we generated 192 monolayer materials and screened eight dynamically stable monolayers through high-throughput calculations. To verify the presence of altermagnetics in these materials, we calculated their band structures and magnetic exchange parameters. Furthermore, we investigated the effects of interchain spacing and external electric fields on altermagnetism of Q1D monolayers.

Our density functional theory (DFT) calculations were carried out using the generalized gradient approximation for the exchange-correlation potential [54], the projector augmented wave method [55], and a plane-wave basis set as implemented in the Vienna Ab-Initio Simulation Package (VASP) [56,57]. In all calculations, the Grimme's D3 form vdW correction was applied to the Perdew Burke Ernzerhof (PBE) exchange functional (PBE-D3) [58]. Kinetic energy cutoffs of 700 eV and 500 eV for the plane wave basis set were used in structural relaxations and electronic calculations, respectively. All atomic positions and lattices were fully relaxed until the residual force per atom was less than 0.001 eV/Å. We used $2 \times 2 \times 1$ supercells for the calculations of structural relaxation and total energy. An $8 \times 2 \times 1$ k-mesh was adopted to sample the Brillouin zone of the $2 \times 2 \times 1$ monolayer supercells. A vacuum layer, more than 15 Å in thickness, was used to reduce interactions among image slabs. On-site Coulomb interactions on d orbitals of the Cr (U = 3.9 eV,J = 1.1 eV) for CrF₃, CrCl₃, CrBr₃, and CrI₃, V (U = 3.0 eV) for VBr₃, Mn (U = 4.9 eV) for MnBr₃, Co (U = 4.9 eV) for CoTe₃, and Fe (U = 3.9 eV) for FeCl₃ were considered using a DFT+U method [59]. The values of U were determined using a linear response method [60] based on the 2D structure with the same chemical formula to the Q1D chains, and their robustness has been tested (Fig. S1 [61]).

Nine structural phases have been experimentally observed and/or theoretical predicted in binary 1D single-atomic magnetic chains, covering four chemical stoichiometric ratios (1:1, 1:2, 1:3, and 1:4) between transition metal (X) and nonmetal (Y) elements [62-65]. Given that altermagnets represent a subset of AFMs requiring zero net magnetization, three categories of magnetic ordering in Q1D monolayers can potentially exhibit altermagnetism: (1) intra- and interchain AFM, (2) intrachain AFM with interchain FM, and (3) intrachain FM with interchain AFM. In the third category, opposite spins are related by a fractional translational symmetry operation, which excludes the possibility of altermagnetism. Therefore, our following analysis specifically targets the first two categories, where magnetic chains exhibit intrachain AFM ordering. These intrachain AFM coupled chains can self-assemble into Q1D monolayers in either AA or AB stacking, with the interchain magnetism being either FM or AFM. Considering variations across stoichiometric ratios, structural phases, interchain stacking orders, and magnetic ordering, we systematically explore 30 Q1D monolayer prototypes, as summarized in Fig. 1 and Figs. S2 to S6 of the Supplemental Materials [61].

Among these 30 prototypes examined, four satisfy the symmetry requirements for altermagnetism, characterized by broken P-T symmetry, as indicated by check marks in Fig. 1(a). These include two γ -phase XY_2 monolayers [prototype 1, Fig. 1(b) and prototype 2, Fig. 1(c)] and two β -phase XY_3 monolayers [prototype 3, Fig. 1(d) and prototype 4, Fig. 1(e)]. Prototypes 1 and 3 adopt AA stacked interchain FM configurations [Figs. 1(b) and 1(c)] while the remaining two are in AB stacked interchain AFM configurations [Figs. 1(d) and 1(e)]. Each prototype structure maintains an

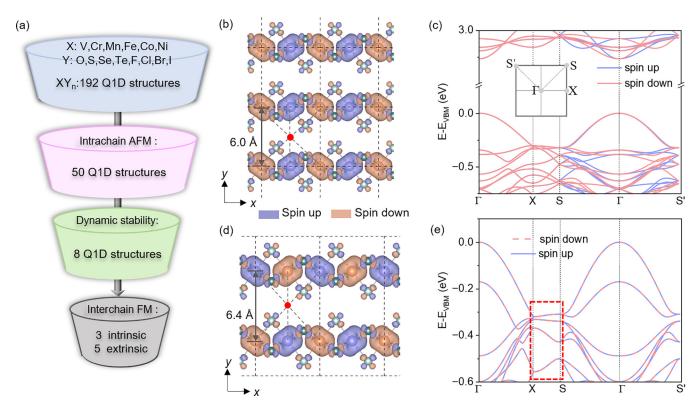


FIG. 2. (a) The screening process of Q1D altermagnets. (b) Top view of spin density distribution and (c) band structure of the $CrCl_3$ monolayer at the interchain spacing of 6.0 Å. The red dot represents the inversion center. The illustration shows the high-symmetry path in the Brillouin zone. (d), (e) The same scheme of (b), (c) for the $CrCl_3$ monolayer with an expanded interchain spacing of 6.40 Å. The red dashed box highlights nodal-line electronic states.

inversion symmetry but lacks the P-T symmetry, which enables the emergence of altermagnetism. Their sublattices with opposite spins are related by the C_{2x} (orange) and M_x (blue) symmetries. As a result, the spin-splitting distribution maps in the Brillouin zone simultaneously reflect these two symmetries, giving rise to an overall d-wave-like (C_{4z}) symmetry [Fig. 1(f)]. Taking prototype 3 [Fig. 1(d)] as an illustrative example, three inversion centers, P1, P2, and P3 (marked by red dots), can be identified. Independent of interchain magnetism, the interchain inversion centers P_1 and P_2 link atoms sharing the same majority spin, thereby inherently breaking the P-T symmetry. Under interchain FM couplings, the interchain inversion center P₃ also links atoms with the same spin, further strengthening the broken P-T symmetry and thereby facilitating the altermagnetism. Conversely, tuning the interchain FM coupling to AFM restores the P-T symmetry, thereby preventing the emergence of altermagnetism.

By realizing all four prototypes through six fourth-period transition-metals and eight chalcogen or halogen nonmetal elements, we constructed 192 Q1D monolayers as potential altermagnet candidates; see Fig. 2(a) blue block. Among them, 50 monolayers consist of single-atomic chains with intrachain AFM ground states (Fig. S7 [61]), suggesting their potential to host altermagnetism [violet block, Fig. 2(a)]. This number further reduces to eight (as listed in Table I) when considering dynamical stability, as verified by their theoretical phonon spectra (Fig. S8 [61]), which show no significant imaginary frequencies [green block, Fig. 2(a)]. Notably, all eight dynamically stable monolayers originate from prototype

3 [Fig. 1(d)]. In contrast, the remaining 42 monolayers show significant imaginary frequencies in their phonon spectra, indicating dynamical instability in their freestanding form.

Interchain FM coupling is presumed in these eight monolayers to host altermagnetism. To clarify the interchain magnetic coupling, we used an Ising model Hamiltonian $H = H_0 - (\frac{J_1}{2} \sum_{ij} S_i \cdot S_j + \frac{J_2}{2} \sum_{\langle ij \rangle} S_i \cdot S_j + \frac{J_3}{2} \sum_{\langle \langle ij \rangle \rangle} S_i \cdot S_j),$ where $ij, \langle ij \rangle$ and $\langle \langle ij \rangle \rangle$ represent the nearest, second-nearest, and third-nearest neighboring sites of magnetic atoms, respectively. S_i represents spin operator at magnetic lattice site *i*. Here J_1 to J_3 denote spin-exchange parameters illustrated in [Fig. 1(e)], among which J_3 specifically represents the interchain spin-exchange coupling. Table I summarizes these parameters for the eight Q1D monolayers. Among them, CrBr₃, VBr₃, and MnBr₃ (Table I) exhibit interchain FM coupling (positive J_3) at their equilibrium interchain distances, indicating that their free-standing monolayers are intrinsic altermagnets [gray block, Fig. 2(a)]. Electronic band structures, obtained from DFT calculations, further verify their altermagnetic characteristics, displaying significant spin splitting along path S- Γ -S' (Fig. S9 [61]), consistent with our symmetry analysis. However, CrF₃ and CrCl₃ have near-zero J_3 values, resulting in nearly degenerate interAFM and FM states. The remaining three Q1D monolayers (CrI₃, FeCl₃, and CoTe₃) exhibit AFM interchain coupling, which prevents them from showing altermagnetic characteristics in their freestanding forms.

Among these eight monolayers, three favor the interchain AFM coupling. To further assess their dynamic stability, we

Monolayer	<i>a</i> (Å)	<i>b</i> (Å)	$J_1 \text{ (meV)}$	$J_2 \text{ (meV)}$	J_3 (meV)	Altermagnet
CrBr ₃	6.32	6.47	-5.83	0.11	0.01	Intrinsic
VBr ₃	6.30	6.59	-21.66	5.53	0.01	Intrinsic
MnBr ₃	6.43	6.39	-1.13	0.43	0.48	Intrinsic
CrF ₃	5.35	4.83	-6.61	0.01	0.00	Extrinsic
CrCl ₃	5.92	6.18	-10.40	-0.01	0.00	Extrinsic
CrI ₃	6.72	7.12	-3.63	0.19	-0.05	Extrinsic
FeCl ₃	5.98	6.28	-1.76	-0.57	-0.19	Extrinsic
CoTe ₃	7.75	4.74	-6.95	-2.26	-0.68	Extrinsic

TABLE I. Lattice constants (*a* and *b*) and spin-exchange parameters [J_1 , J_2 , J_3 , labeled in Fig. 1(e), in units of meV per magnetic atom] of the eight dynamically stable AA-stacked intrachain AFM β -XY₃ Q1D monolayers.

examined vibrational frequencies of interchain AFM coupled structures. The absence of significant imaginary frequencies at the Γ point, along with the nonimaginary frequencies observed in their interchain FM coupled counterparts, demonstrates the feasibility of tuning interchain magnetic coupling between AFM and FM in these monolayers. The CrCl₃ Q1D monolayer, composed of experimentally synthesizable CrCl₃ single-atomic chains, exhibits nearly degenerate interchain FM and AFM couplings in its freestanding form. Modulating the interchain spacing at vdW gaps effectively tunes the interchain magnetic coupling [66], which plays a paramount role in the emergence of altermagnetic characteristics. A lower interchain force constant facilitates this tuning process. Notably, the interchain force constant of 5.06×10^{19} N/m³ for monolayer CrCl₃ is only half to the interlayer value for black phosphorus $(10.1 \times 10^{19} \text{ N/m}^3)$ [67]. This soft interchain modulus enables feasible tuning of the interchain spacing from the equilibrium lattice constant b, 6.18 Å theoretically.

When the spacing is reduced to 6.00 Å, J_3 increases to 0.02 meV/Cr, favoring an altermagnetic state. At this distance, the spin density distribution of the CrCl₃ monolayer illustrates interchain aligned local magnetic moments (the FM state) [Fig. 2(b)]. Although it possesses inversion symmetry, the P-T symmetry has been broken. Band structure calculations verify the expected spin splitting along the S- Γ -S' path [Fig. 2(c)]. On the other side, as the interchain spacing expands to 6.40 Å, the nearly zero J_3 becomes negative and reaches -0.01 meV/Cr, stabilizing an interchain AFM state [Fig. 2(d)]. The two sublattices preserve the inversion and P-T symmetries simultaneously. This configuration corresponds to a Néel AFM semiconductor with a bandgap of 2.84 eV. Our band structure calculations further reveal symmetry-protected fourfold degenerate nodal lines along the X-S direction [Fig. 2(e)] [68]. These results illustrate the tunability of interchain spacing for introducing transitions between an altermagnetic semiconductor and a Néel antiferromagnetic semiconductor.

Two distinct trends emerge in magnetic-spacing relations among these eight monolayers. One such trend, exemplified by the VBr₃ monolayer, indicates that decreasing interchain spacing (lattice constant **b**) favors the FM coupling [Trend D, Fig. 3(a)]. This behavior is also observed in CrBr₃, CrI₃, CrCl₃, and MnBr₃ monolayers (Fig. S10 [61]). When the interchain spacing of the VBr₃ monolayer increases from its equilibrium value of 6.59 Å to 6.80 Å [red pentagram in Fig. 3(a)], the interchain magnetism becomes AFM from the original FM. The AFM coupling leads to a transition of the monolayer from an altermagnet to a nodal-line semiconductor with a bandgap of 1.65 eV, as shown in Fig. 3(b).

The CoTe₃ monolayer follows an opposite trend, where increasing the interchain spacing prefers the interchain FM coupling (Trend I). As shown in Fig. 3(c), a gradual increase in the interchain spacing transitions the initially favored interchain AFM coupling to FM. The interchain spin-exchange parameter J_3 reduces from -0.68 meV/Co to zero and subsequently increases to 1.09 meV/Co (Table I) as the interchain spacing expands from its equilibrium (4.74 Å) to 5.10 Å [red pentagram in Fig. 3(c)]. At the 5.10 Å spacing, significant spin splitting is observed in the valence band along the S- Γ -S' path [Fig. 3(d)]. This trend, where increasing the interchain spacing weakens the interchain AFM coupling, is also consistently observed in CrF₃ and FeCl₃ monolayers (Fig. S11 [61]).

The interchain magnetic ground state is determined by the competition between Pauli repulsion and interchain hopping [66,69]. Unlike the 2D case, Q1D monolayers exhibit two distinct interchain bonding states that result in Pauli repulsion at vdW gaps, resulting in double antiferromagnetic regions in the Bathe-Slater curve (Fig. S12 [60]). This two-orbital Bathe-Slater curve features two Trend I regions and one Trend D region. The equilibrium interchain distances of CrCl₃, VBr₃, CrI₃, CrBr₃, and MnBr₃ fall within the Trend D region, while those of CrF₃, FeCl₃, and CoTe₃ lie in the Trend I region. Among these eight Q1D monolayers, CoTe₃ exhibits the largest spin splitting, reaching several hundred meV, while CrF₃ and MnBr₃ show the largest (4.56 eV) and smallest (0.19 eV) bandgaps, respectively (Table S1). Recently experimental studies demonstrate that hydrostatic pressure [70], nanotube radius variation [71], and coverage adjustment [72] can tune interchain distances in 1D systems. We also infer that substrate-mediated tuning can be achieved via vdW epitaxy. These experimental achievements, together with those results discussed earlier, establish interchain spacing as an effective means for tailoring electronic and magnetic properties in Q1D monolayer altermagnets.

We next examined the effect of electric fields on altermagnetism, using monolayer $CrCl_3$ as a representative. Due to its mirror symmetry along the *z* axis, monolayer $CrCl_3$ exhibits identical response behaviors under both positive and negative electric fields. Therefore, we focus on the effects of a positive electric field in the following discussion. In the interchain FM configuration, the valence bands of the freestanding $CrCl_3$ monolayer, mainly composed of Cl-p orbitals (Fig. S13

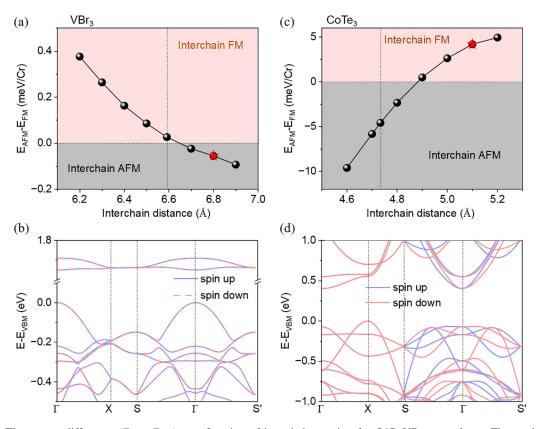


FIG. 3. (a) The energy difference $(E_{AFM}-E_{FM})$ as a function of interchain spacing for Q1D VBr₃ monolayer. The vertical dashed line indicates the freestanding interchain distance. (b) Band structure of the monolayer VBr₃ under interchain of 6.80 Å [labeled as red pentagram in 3(a)]. (c) The energy difference as a function of interchain spacing for Q1D CoTe₃ monolayer. (d) Band structure of the monolayer CoTe₃ under interchain of 5.10 Å [labeled as red pentagram in 3(c)].

[61]), show relatively weak spin splitting (~10 meV) along the S- Γ -S' path. The splitting nearly doubles (~20 meV) at an electric field of 0.2 V/Å [Fig. 4(a)]. An accidental band crossing between spin-up and spin-down components occurs in the valence band at 0 V/Å, which was captured by the band structure plot along path Γ -S [orange dot in Figs. 4(a) and 4(b)]. This crossing leads to spin-component changes, and when combined with the preserved C_{2x} rotation and M_x mirror symmetries, the map exhibits an eightfold pattern in the momentum-resolved spin splitting distribution. Upon applying an out-of-plane electric field of 0.2 V/Å, the broken inversion symmetry enlarges the spin splitting, which avoids the accidental crossing spanning the first Brillouin zone, thereby reducing the spin splitting pattern from eightfold to fourfold.

For the interchain AFM configuration of the CrCl₃ monolayer, while the electric field breaks the P-T symmetry, a fractional translational symmetry of (0, 1/2, 0) prevents the emergence of altermagnetism. However, in certain structures, interchain sliding [CrF₃, Fig. S14(a) [61]] or rotating [CoTe₃,

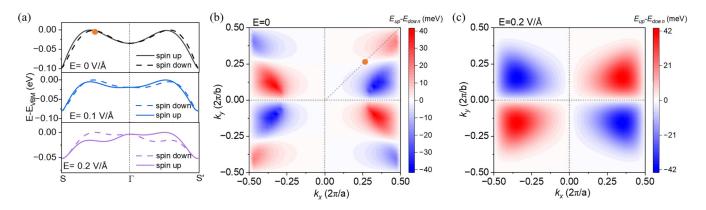


FIG. 4. (a) Band dispersion plots of the highest valence band in freestanding $CrCl_3$ monolayer with interchain FM coupling under varied external electric field. The orange dots indicate the band crossing point along the Γ -S direction. Spin splitting mappings of the highest valence band in the freestanding $CrCl_3$ monolayer (b) without electric field and (c) under an electric field of 0.2 V/Å.

Fig. S14(c) [61]] occurs in interchain AFM configurations (Fig. S14 [61]). In these cases, an external electric field breaks both the P-T and the fractional translational symmetries, inducing spin splitting. Although the magnitude of spin splitting in these structures is relatively small (less than 2 meV), this finding demonstrates the rich tunability inherent in Q1D monolayers. These results highlight the flexibility and tunability of altermagnetism in Q1D monolayers under an external electric field.

In summary, we extended the family of altermagnetic materials to Q1D single-atomic magnetic chain structures using a self-assembly strategy. Through symmetry analysis and highthroughput calculations, we predicted eight stable Q1D monolayer, including three intrinsic and five tunable altermagnets, assembled from AA-stacked and intrachain AFM coupled β -XY₃ single-atomic magnetic chains. These monolayers have highly anisotropic interatomic interactions (covalent versus vdW interactions) and electronic structures. They also exhibit various band gaps and tunable spin splitting. Our study highlights the feasible tunability and the potential for artificial engineering of altermagnetism through interchain spacing in these highly anisotropic Q1D monolayers assembled from single-atomic chains. Specifically, we identified two distinct trends. In Trend I, increasing interchain spacing stabilizes FM interchain coupling in monolayers like CoTe₃, leading to enhanced spin splitting. In Trend D, a decrease in interchain spacing strengthens altermagnetism in monolayers like VBr₃. Moreover, external electric fields provide an additional degree of control over spin splitting in altermagnetism, demonstrating the flexibility of Q1D altermagnets for potential device integration. While our study establishes the dynamical stability of AA-stacked β -XY₃ altermagnets, their thermodynamic stability warrants further investigations [77]. Additionally, although free-standing γ -XY₂ monolayers are dynamically unstable, stabilization strategies, such as substrate engineering or charge doping, present promising avenues for exploration. This work broadens the scope of low-dimensional altermagnets and provides a blueprint for designing novel spintronic materials with tunable magnetic and electronic properties.

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Data availability. The data that support the findings of this article are not publicly available upon publication because it is not technically feasible and/or the cost of preparing, depositing, and hosting the data would be prohibitive within the terms of this research project. The data are available from the authors upon reasonable request.

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